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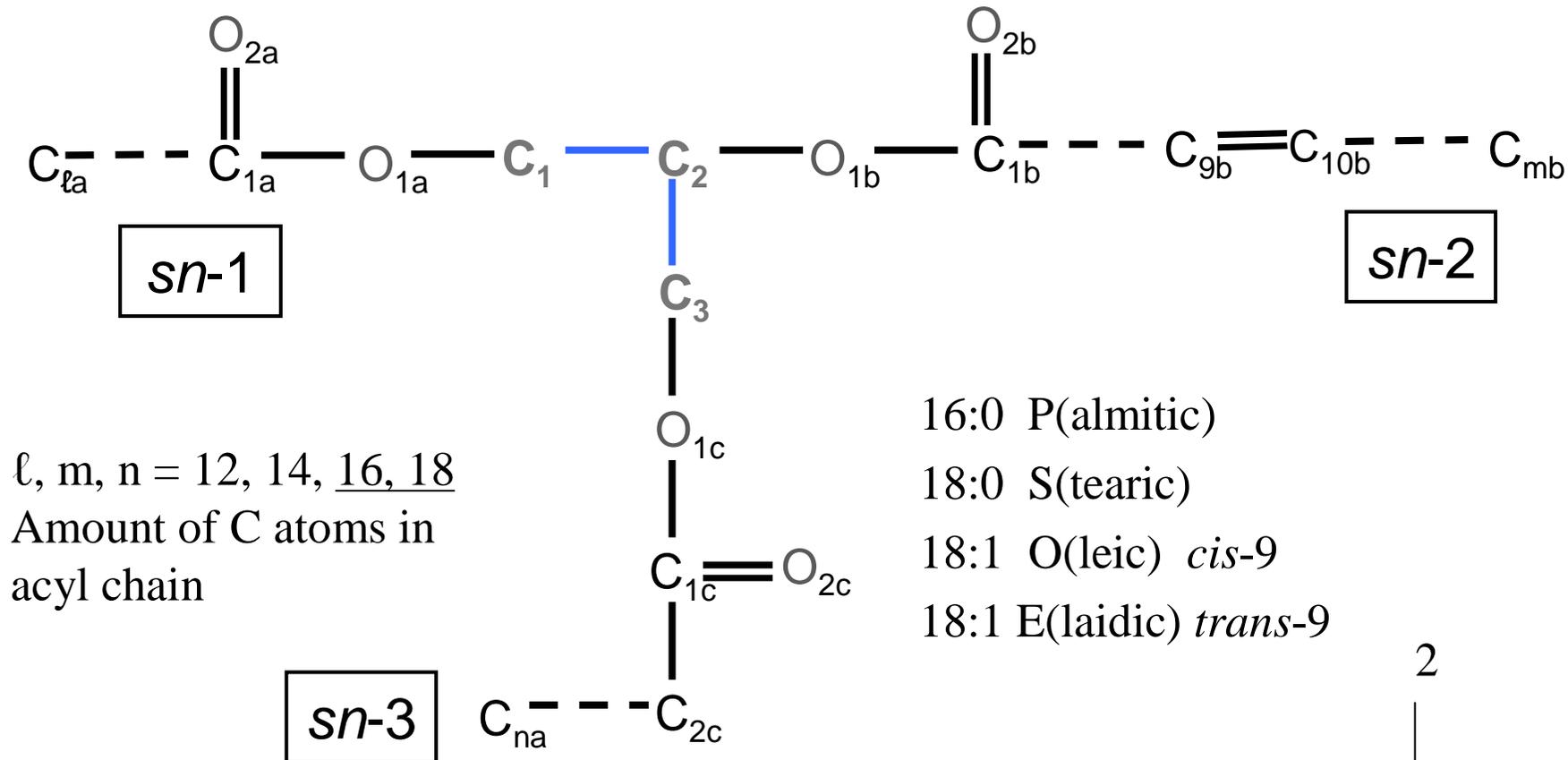
# Structure and polymorphism of *trans* mono-unsaturated triacylglycerols and related fully saturated TAGs

**Rene Peschar**

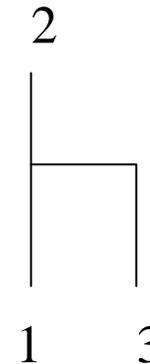
Jan B. van Mechelen, Henk Schenk

- Introduction
- Time-resolved X-ray powder diffraction (XRPD)
  - Phase transitions and stability of TAG polymorphs
- Structure determination from XRPD data
  - $\beta$ -2 polymorphs
  - $\beta'$ -2 polymorphs
- Conclusions

# Introduction. Triacylglycerol (TAG) conformations

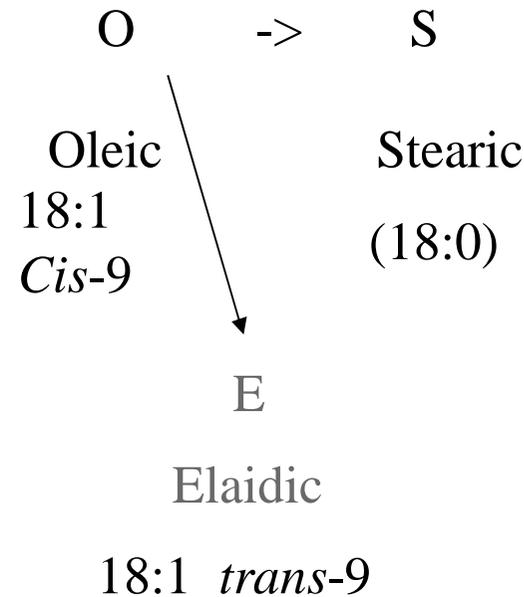
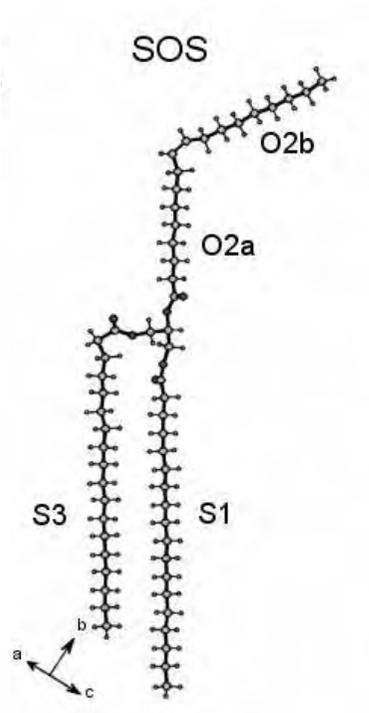


e.g. [1-3] conformation *sn-1* and *sn-3* in the same direction



# Introduction. Trans fatty acid

Hydrogenation changes cis un-saturated fatty acid chains partly into *trans* isomers



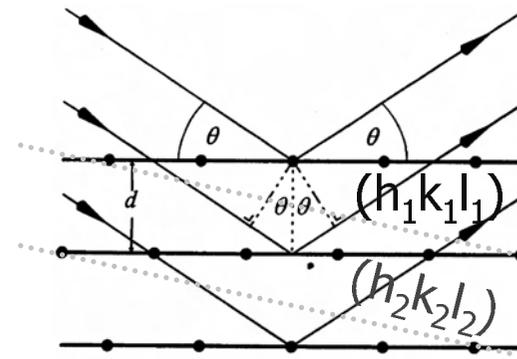
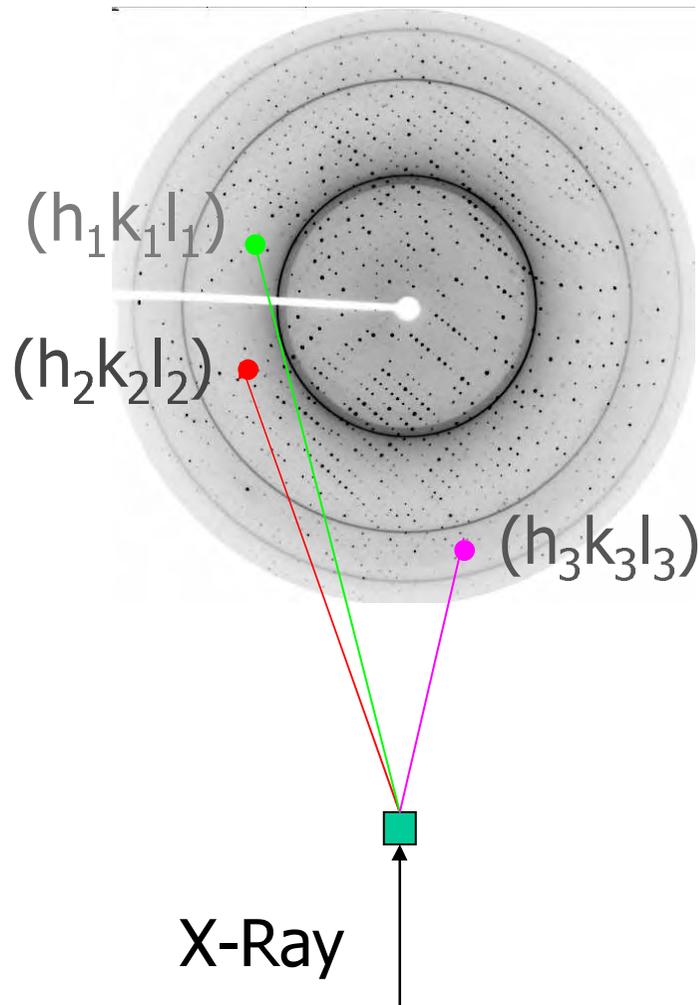
[1-3] conformation

- What is the influence of the *trans* fatty acid elaidic acid on the packing and stability of TAG polymorphs ??
- To which extent can E replace an S?

# Introduction. X-ray diffraction

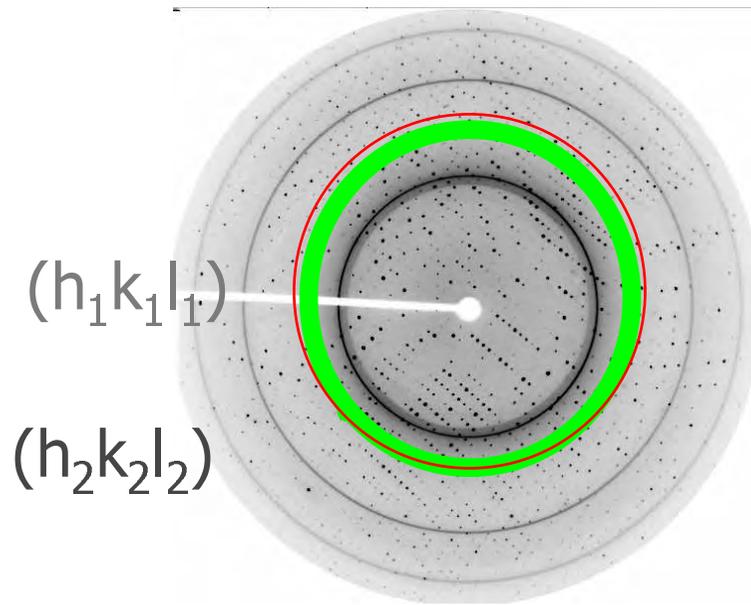
- Crystal: regular 3D stacking of identical units

- X-rays on crystal => diffraction  
Bragg's Law :  $2d_{hkl}\sin(\theta) = \lambda$



- $(h k l)$  lattice plane or reflection

# Introduction X-ray powder diffraction



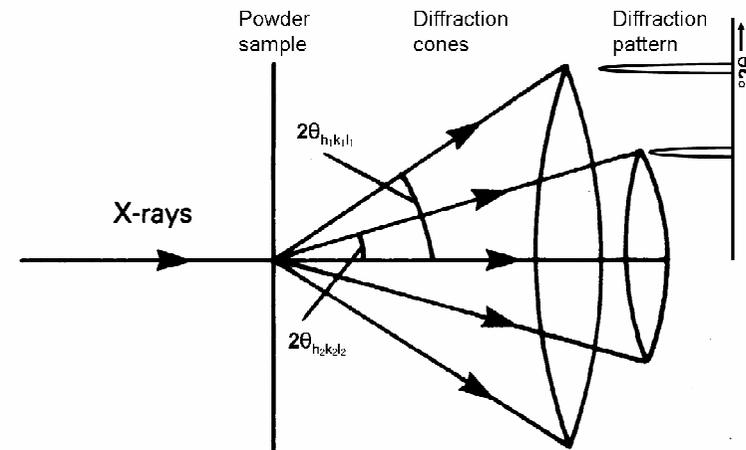
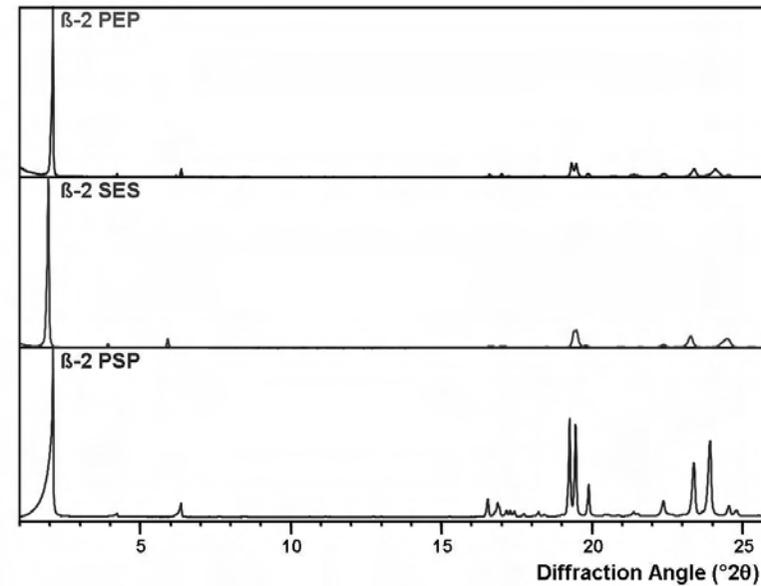
X-Rays

Powder = large set of small crystals  
( $< 10 \mu\text{m}$ )

Reflections from different crystal  
planes overlap

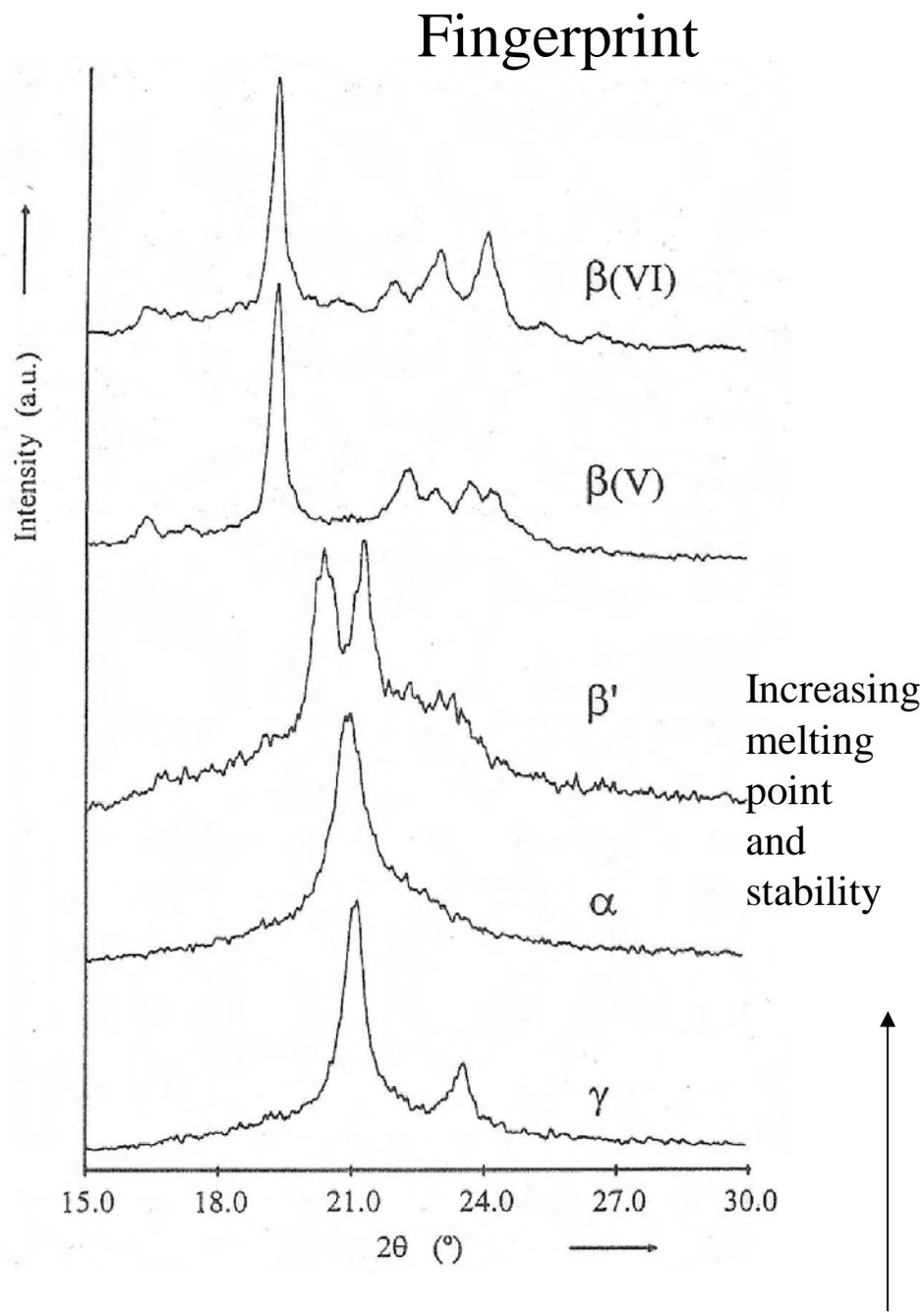
Low angle

Fingerprint



# XRPD polymorph identification

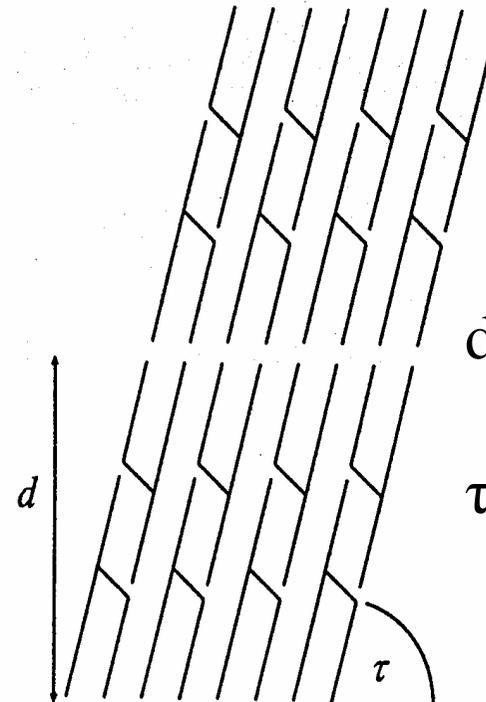
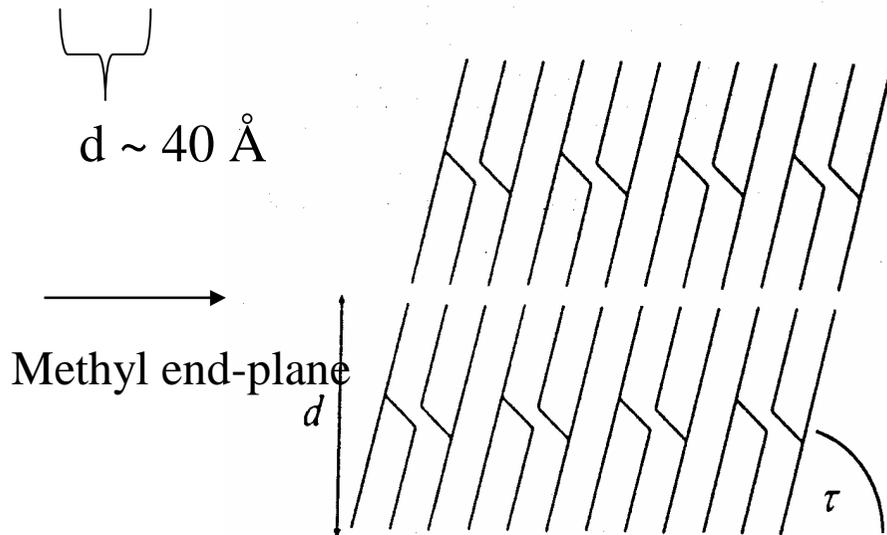
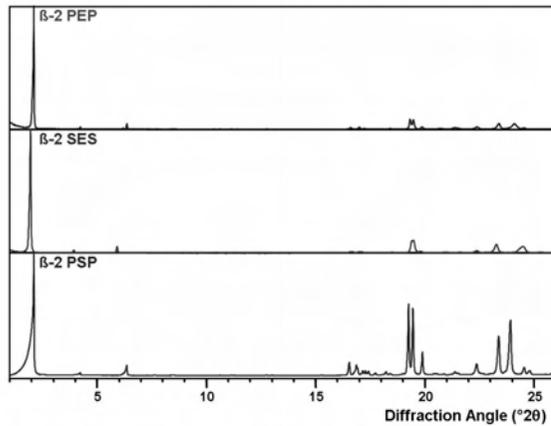
Polymorph	Long spacing (Å)	Short spacings (Å)		
$\gamma$	52.4		4.14	3.69
$\alpha$	48.4		4.18	
$\beta' - 2$	45.0		4.33	4.14
$\beta - V = \beta_{2-3}$	64.8	4.58	3.99	3.87 3.76 3.67
$\beta - VI = \beta_{1-3}$	64.4	4.58	4.03	3.85 3.69



# XRPD $\beta'$ and $\beta$ chain-length packing

Left: Double chain-length seat-facing chairs (e.g.  $\beta$ -2)

Right: Triple chain-length packing (e.g.  $\beta$ -3)



$d$ : Long d-spacing  
(70  $\text{\AA}$ )

$\tau$ : Tilt angle

# Polymorphs of investigated TAGs

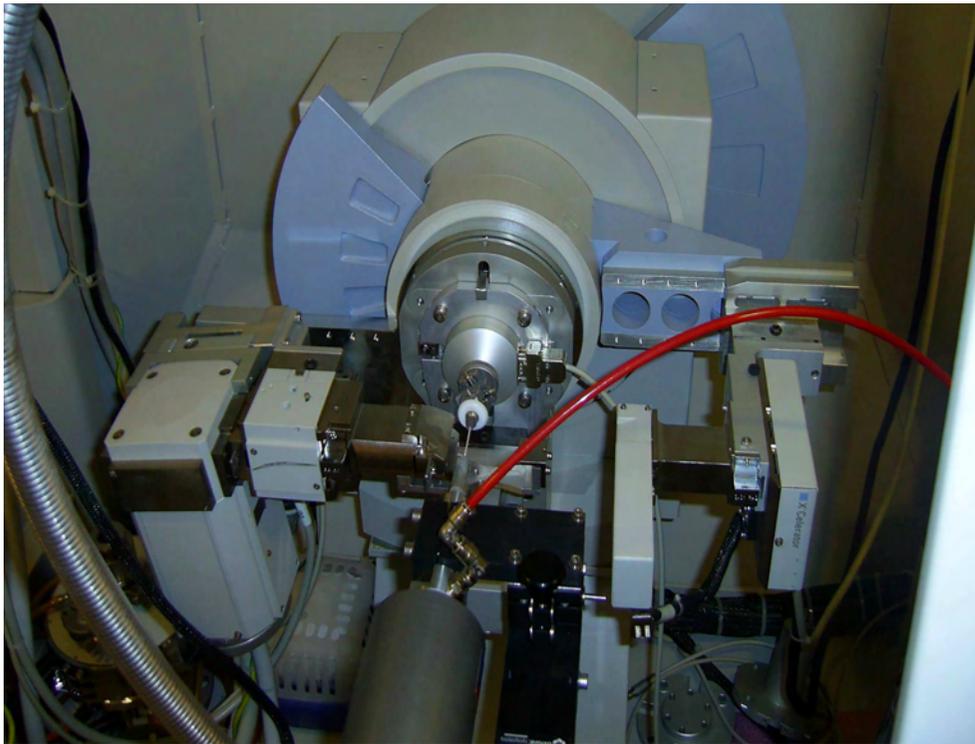
increasing stability



TAG	$\alpha$	$\beta'_{2-2}$	$\beta'_{1-2}$	$\beta-2$	
PEP		MS	Stable	Exists, but lower melting!	
PSP		MS	Stable	Exists, but lower melting!	
PPE		MS	MS	Stable	
PPS		MS	MS	Stable	MS: Metastable
PSS		MS	MS	Stable	Not reported in literature

- Are these all the possible polymorphs?
- What is the difference between  $\beta'_{2-2}$  and  $\beta'_{1-2}$ ?
- PEP and PSP are  $\beta'$  stable but a lower melting  $\beta$  exists?
- What is the packing of the polymorphs ?
- What is the influence of replacing E by S?

# Time- and temperature resolved XRPD PANalytical X'pert Pro MPD



- Elliptical focusing mirror
- Soller slits (prim., sec. 0.02 rad div)
- X'celerator strip detector
- Oxford Instruments Cryostream Compact

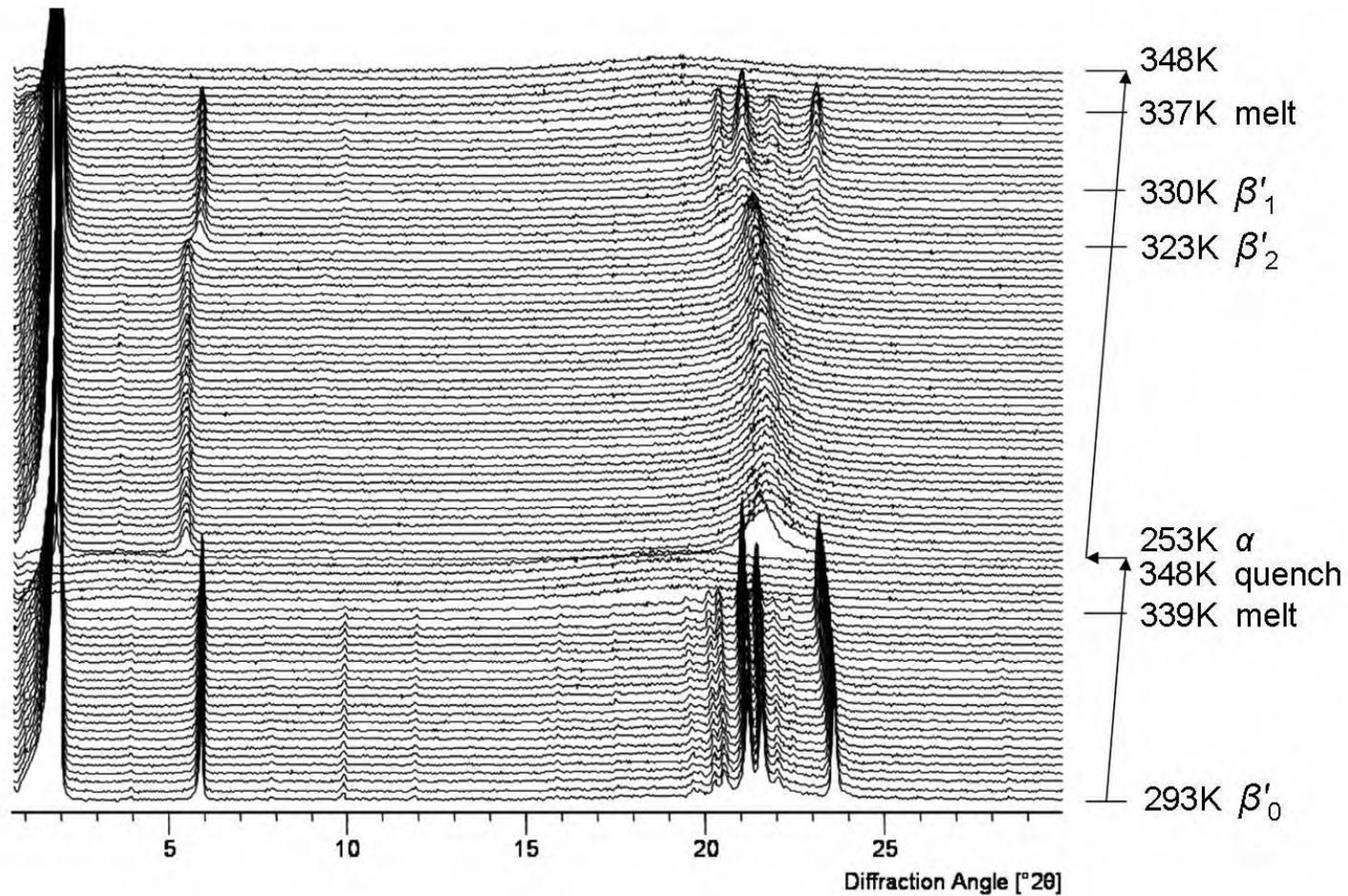
- N<sub>2</sub> stream parallel to capillary
- Cylindrical polymer film

Cu K $\alpha$ , high flux

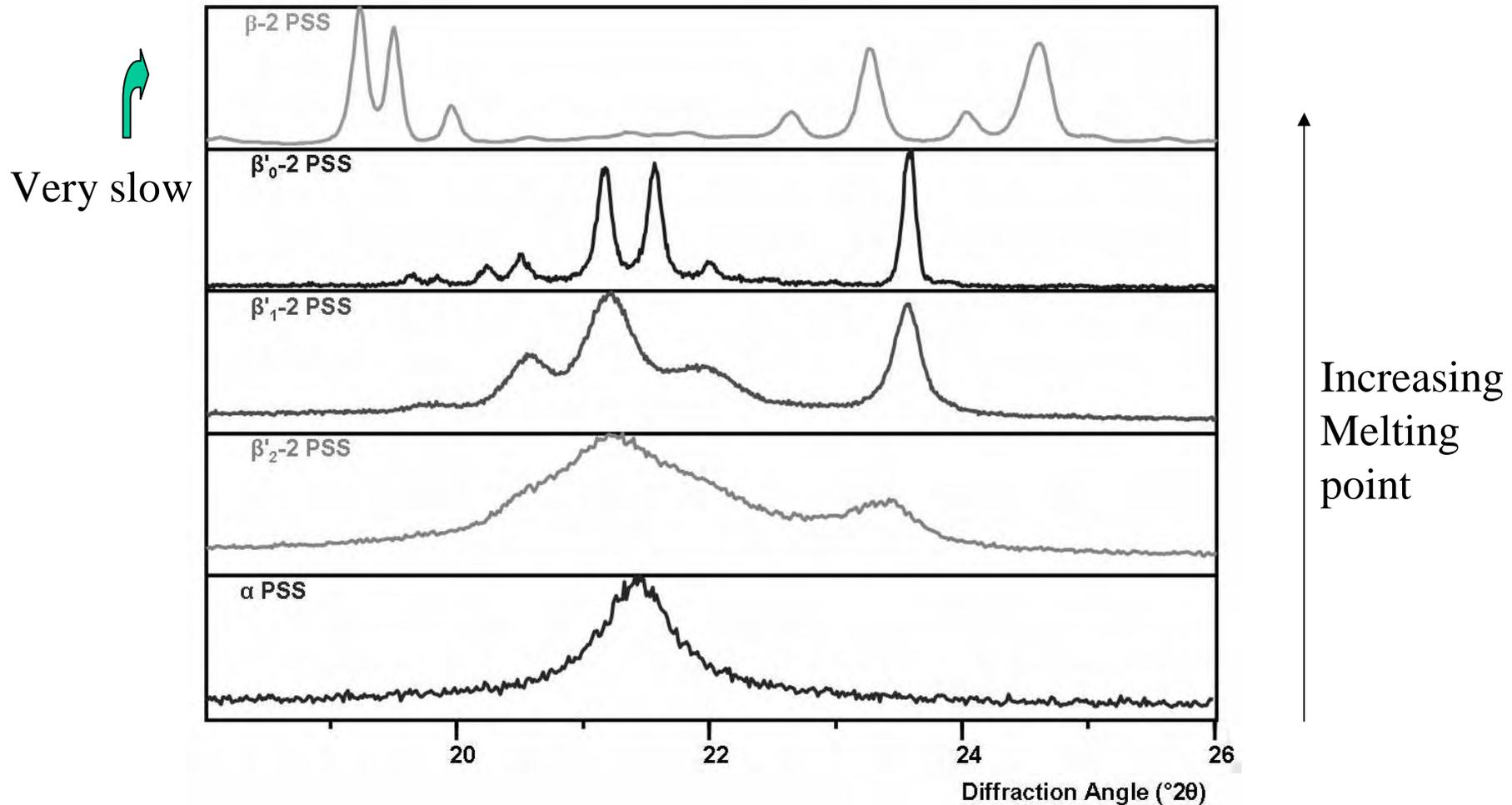
Typical scan settings: 1 min. 0.5-30 °2 $\theta$  step 0.016 °2 $\theta$   
Heating and cooling rates: 0.5 – 6/30 K min<sup>-1</sup>

# Melt and crystallization of polymorphs of PSS

quenching ( $30 \text{ K min}^{-1}$ )  
heating ( $0.5 \text{ K min}^{-1}$ )



# Fingerprint area of polymorphs of PSS



- $\beta'_1$ -2 is a higher crystalline form of  $\beta'_2$ -2
- $\beta'_0$ -2 is a novel polymorph

# Phase transitions and stability of polymorphs

Melting points and transition temperatures in K

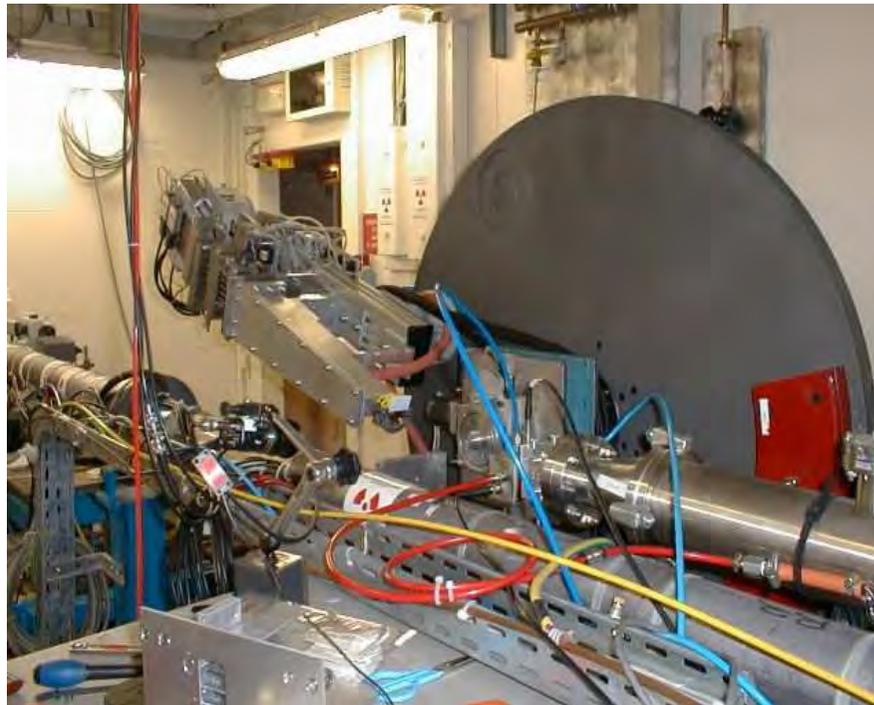
Heating rate 0.5 Kmin<sup>-1</sup>

	Transition	transition	melt	melt	melt	
TAG	$\alpha$ to $\beta'_{2-2}$	$\beta'_{2-2}$ to $\beta'_{1-2}$	$\beta'_{1-2}$	$\beta'_{0-2}$	$\beta-2$	
PEP	303	312	329	-	327	} $\beta'$ stable
PSP	317	319	343	-	339.5	
PPE	304	311	320	-	320	
PPS	321	323	332	-	338	} $\beta$ stable
PSS	323	330	337	339	339	

•E stabilizes  $\beta'_{2-2}$   
 •Asymmetric  $\beta'_{2-2}$  TAGs melt faster  
 •No significant difference E vs. S

# XRPD data collection for structure determination European Synchrotron Radiation Facility (ESRF) Grenoble France

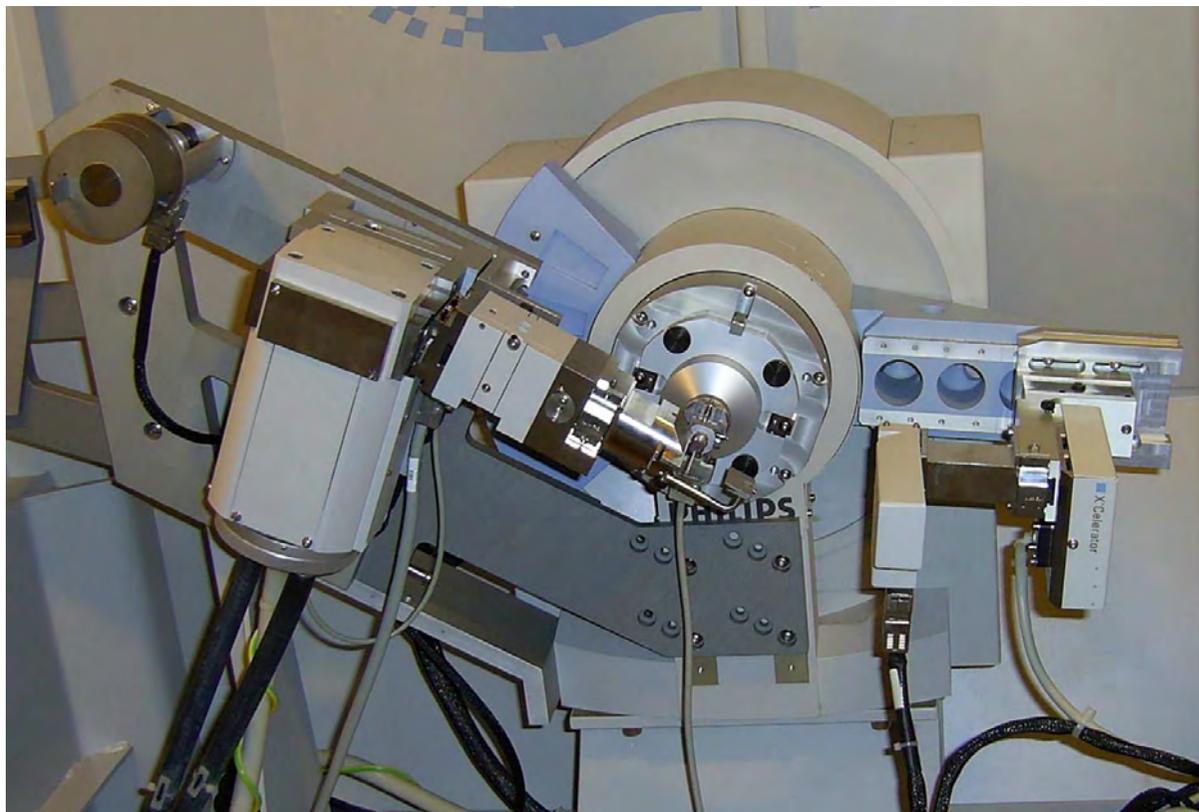
Beamline BM01B



- 2 circle diffractometer
- 6 detectors => 6 complete patterns simultaneously
- Si 111 analyzer crystal in front of each detector =>  
FWHM  $\sim 0.01^\circ 2\theta$  ( $\lambda = 1.0 \text{ \AA}$ )
- $\lambda = 0.8 \text{ \AA}$

# XRPD data collection for structure determination

## PANalytical X'pert Pro Alpha1



- Hybrid monochromator
- Soller slits (prim., sec. 0.01 rad div)
- X'celerator strip detector
- Oxford Instruments Cryostream Compact

Cu  $K\alpha_1$       $\lambda = 1.5406 \text{ \AA}$

$N_2$  stream parallel to capillary  
Cylindrical polymer film

Typical settings:  $0.5\text{-}30^\circ 2\theta$  step  $0.008^\circ 2\theta$

## Structure determination from XRPD data

1. Sample preparation (Capillary 0.7 mm)
2. Data collection
3. Find peak positions
4. Unit cell determination (indexing)
5. Direct space search
6. (Rietveld) refinement

## Indexing; limit indexing search space

Make guess of volume asymmetric unit:

Specific weight } Use info from similar crystal structures  
to set limits to cell dimensions  
Molecular weight } **Volume / molecule**  
**Z x Volume asymmetric unit = Unit cell volume**

Generally only 1<sup>st</sup> 20 lines are used for indexing

- De Wolff FOM:

$$M_{20} = Q_{20} / (2 \cdot \langle Q \rangle \cdot N_{20}) \quad M_{20} > 10$$

$\langle Q \rangle$  = average  $|Q_{\text{calc}} - Q_{\text{obs}}|$

$N_{20}$  = number of calc. Q's upto  $Q_{20}$

$\theta_g$  = selected limit

$N(\theta_g)$  = number of calc.  $\theta$ 's upto  $\theta_g$

$\langle 2\theta \rangle$  = average discrepancy     $N$  = number of observed lines upto  $\theta_g$

• Do not accept cells with unexplained non-indexed lines

## Examples of indexing software

- ITO zone finding  
needs accurate low angle data
- TREOR trial and error + experience  
needs accurate low angle data
- DICVOL dichotomy method  
less error sensitive
- McMaille Monte Carlo , grid search
- Index (slow)
- Kohl low symmetry cells (DOS)
- Lzon low symmetry cells (DOS)
- LSQDETC Grid Search
- FOX beta version

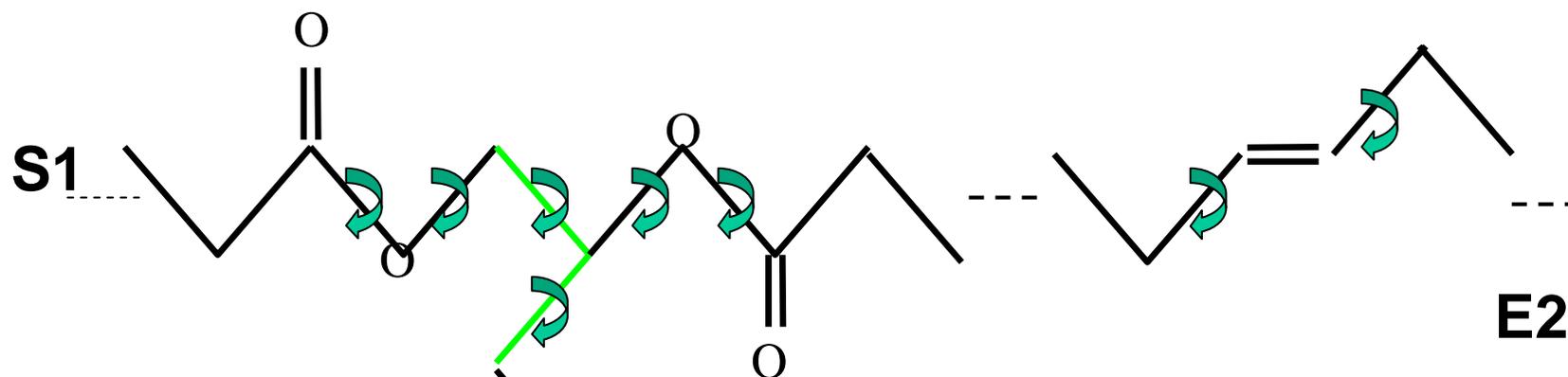
$M_{20}$  alone is not sufficient

Check if complete pattern is covered (e.g CHEKCELL)

# Structure determination from XRPD data

Direct space search

Fox, simulated annealing



Molecule model: Z-matrix  
Connected rigid (planar) chains

S3

Free rotation and translation  
Selected free torsion angles

Favre-Nicolin, V. & Černý, R. (2002). *J. Appl. Cryst.* **35**, 734-743.

## Crystal structures of polymorphs solved

Melting points and transition temperatures in K

	Transition	transition	melt	melt	melt
	$\alpha$ to $\beta'_{2-2}$	$\beta'_{2-2}$ to $\beta'_{1-2}$	$\beta'_{1-2}$	$\beta'_{0-2}$	$\beta-2$
• PEP	303	312	329	-	327
• PSP	317	319	343	-	339.5
• PPE	304	311	320	-	320
• PPS	321	323	332	-	338
• PSS	323	330	337	339	339

Crystal structure determined

Novel polymorph, crystal structure determined

# XRPD patterns of $\beta$ -2 polymorphs of symmetric TAGs

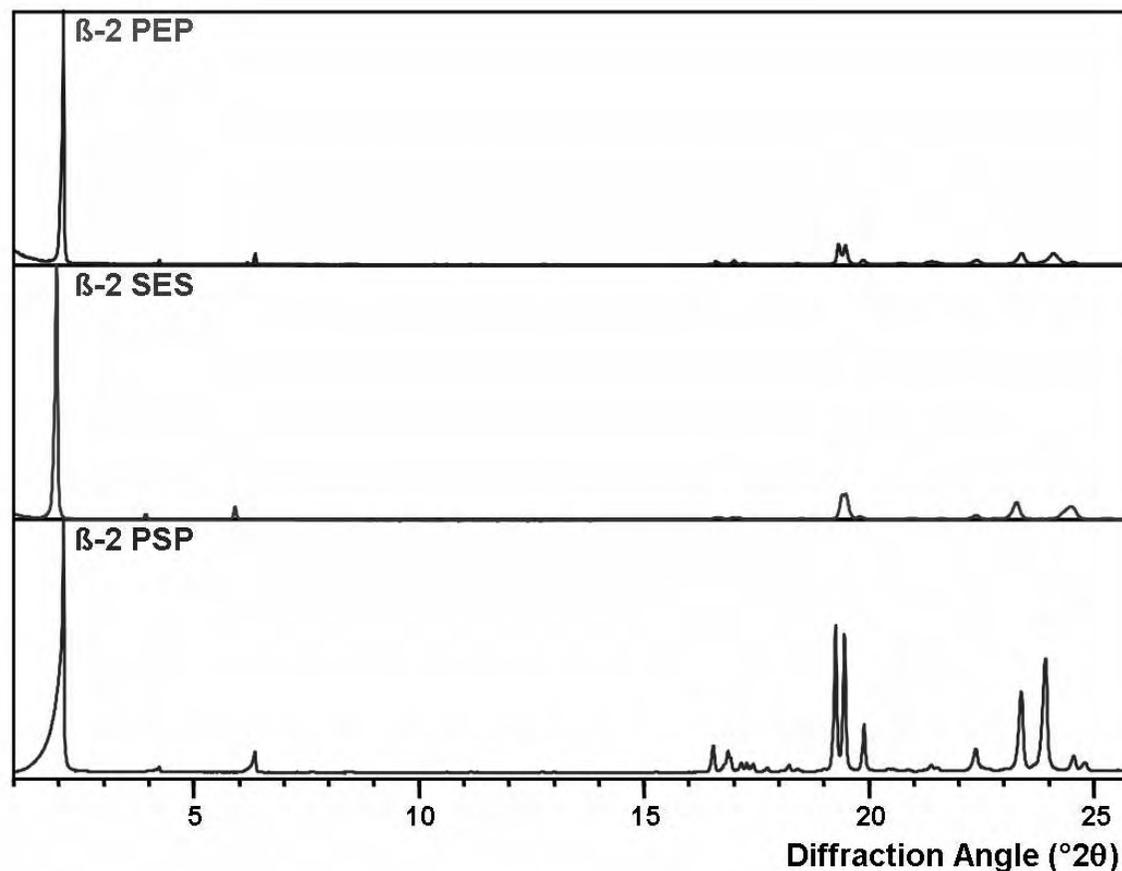
a, b, c in Å  
 $\alpha, \beta, \gamma$  in °

5.417 12.13 41.54  
 88.3 92.09 99.95

5.425 11.95 44.64  
 92.01 88.76 100.18

5.439 12.18 41.60  
 88.73 93.10 99.97

Space group P-1



BM1b

BM1b

BM16

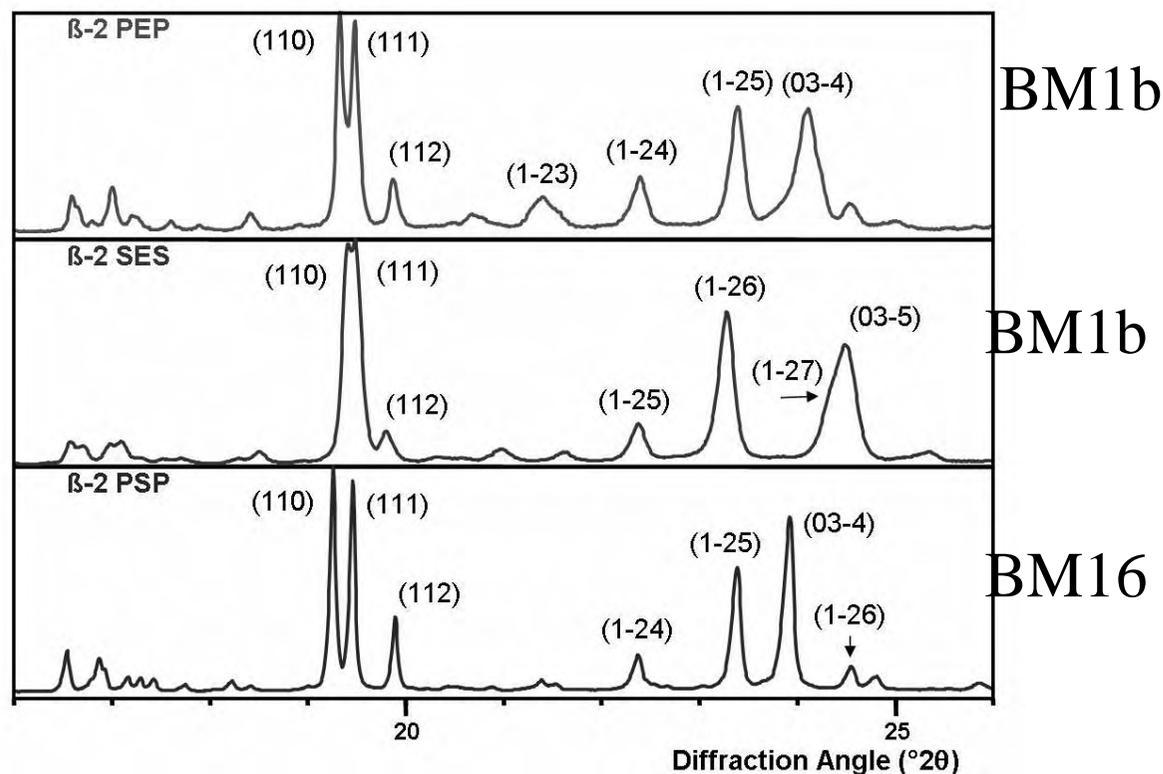
(0 0  $\ell$ )

All data scaled to  
 CuK $\alpha$ 1

•Van Mechelen et al. (2008) Acta Cryst B64, 240-248.

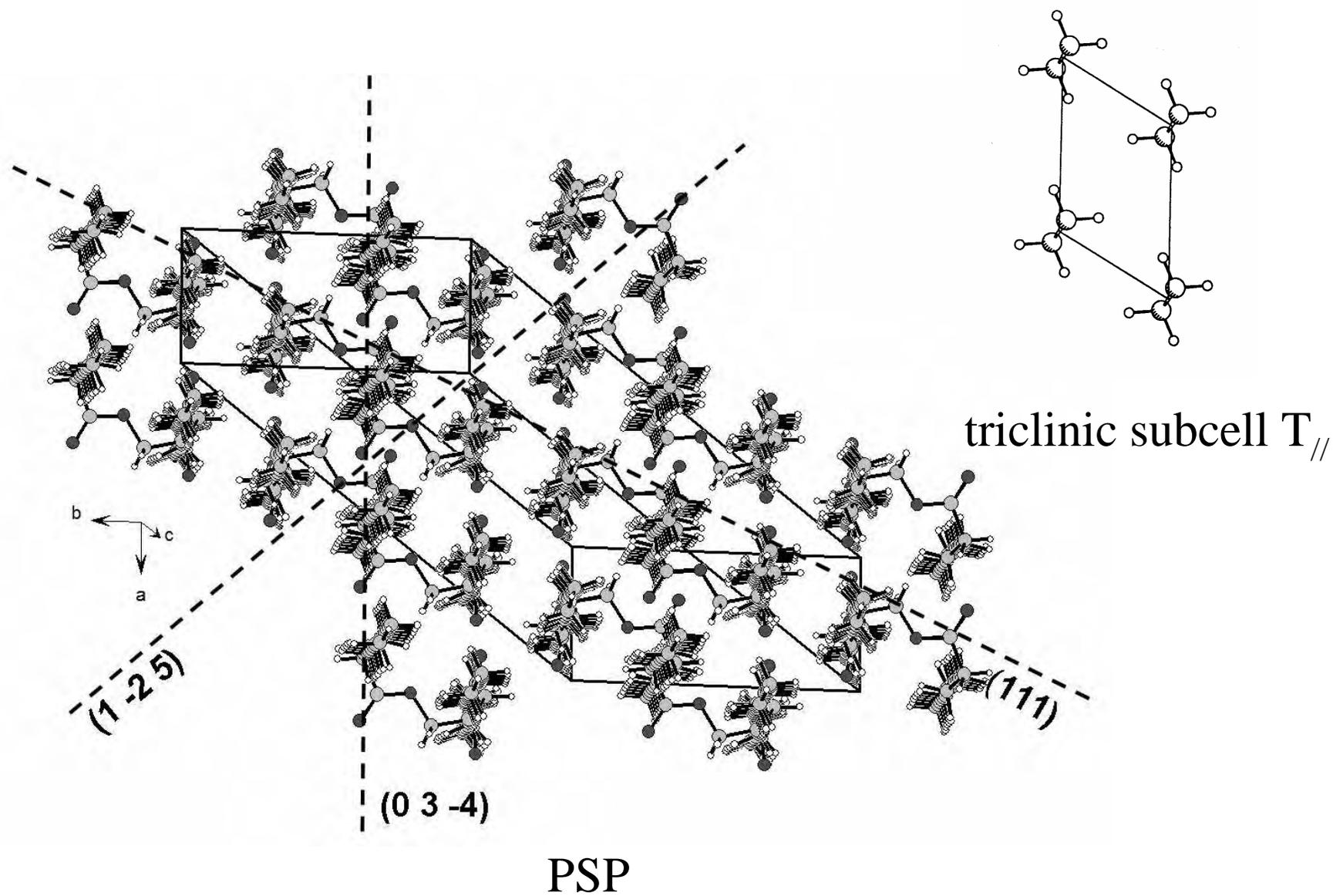
## XRPD fingerprint area of $\beta$ -2 polymorphs of symmetric TAGs

Polymorph	Short spacings (Å)		
$\gamma$		4.14	3.69
$\alpha$		4.18	
$\beta' - 2$		4.33	4.14
$\beta_2 - 3$	4.58	3.99	3.87
			3.76
			3.67
			3.69
$\beta_1 - 3$	4.58	4.03	3.85
			3.69



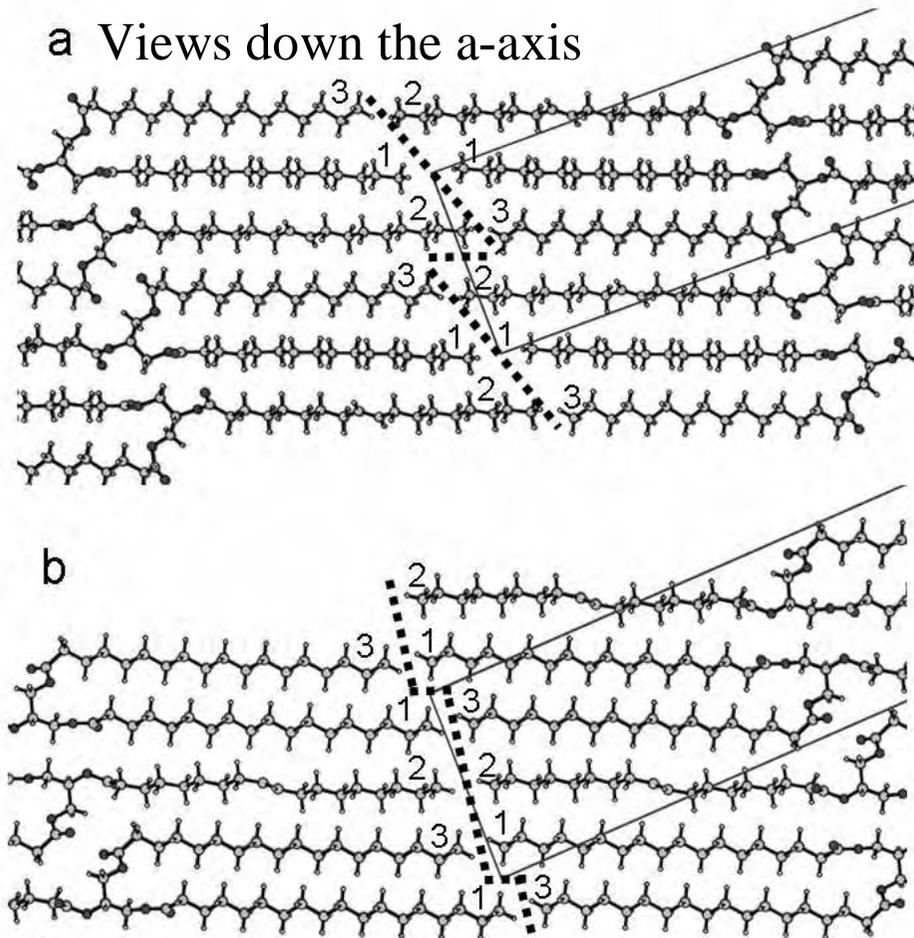
Dominant reflections: (111) and (0 -3 4)

# $\beta$ -2 polymorphs dominant reflections



# Packing characteristics of the $\beta$ -2 polymorphs

- Chair-shaped [1-3] conformation, seat-facing (inversion centres) (same as in e.g.  $\beta$ -2 SSS)
- Pairs of chairs form 'two-packs' : layer with double-chain thickness
- Two-packs face each other (shifted over  $a/4$ ) at the methyl end-plane



PEP

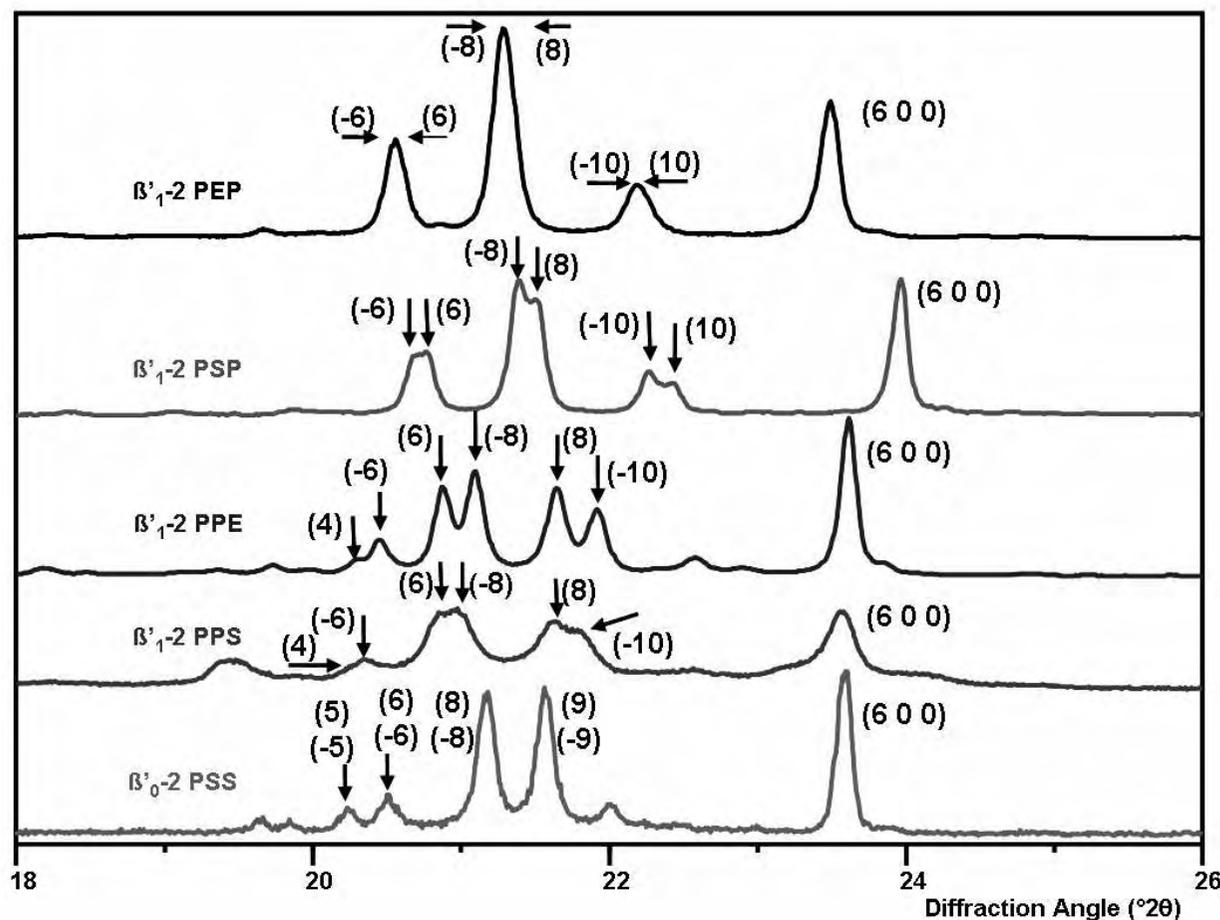
- Chain directions (1-1) aligned
- Two-packs tilted
- Small angle between step plane (- - -) and chain direction (1-1)

SES

- Chain directions (2-2) aligned
- Two-packs not tilted
- Large angle between step plane (- - -) and chain directions (2-2)

No significant difference in packing for E vs S !

# Fingerprint area of $\beta'_1$ -2 PEP, PSP, PPE, PPS and $\beta'_0$ -2 PSS



Cell parameters Spacegroup

$\beta'_1$ -2 PEP	22.715	5.656	85.110	I2
		90.2		
$\beta'_1$ -2 PSP	22.253	5.634	85.263	I2
		90.8		
$\beta'_1$ -2 PPE	22.988	5.641	86.265	I2
		93.52		
$\beta'_1$ -2 PPS	22.7515	5.650	86.746	I2
		93.97		
$\beta'_0$ -2 PSS	22.651	5.653	89.462	C2/c
		90.01		

Only the  $\ell$  of the  $(31\ell)$  is shown.

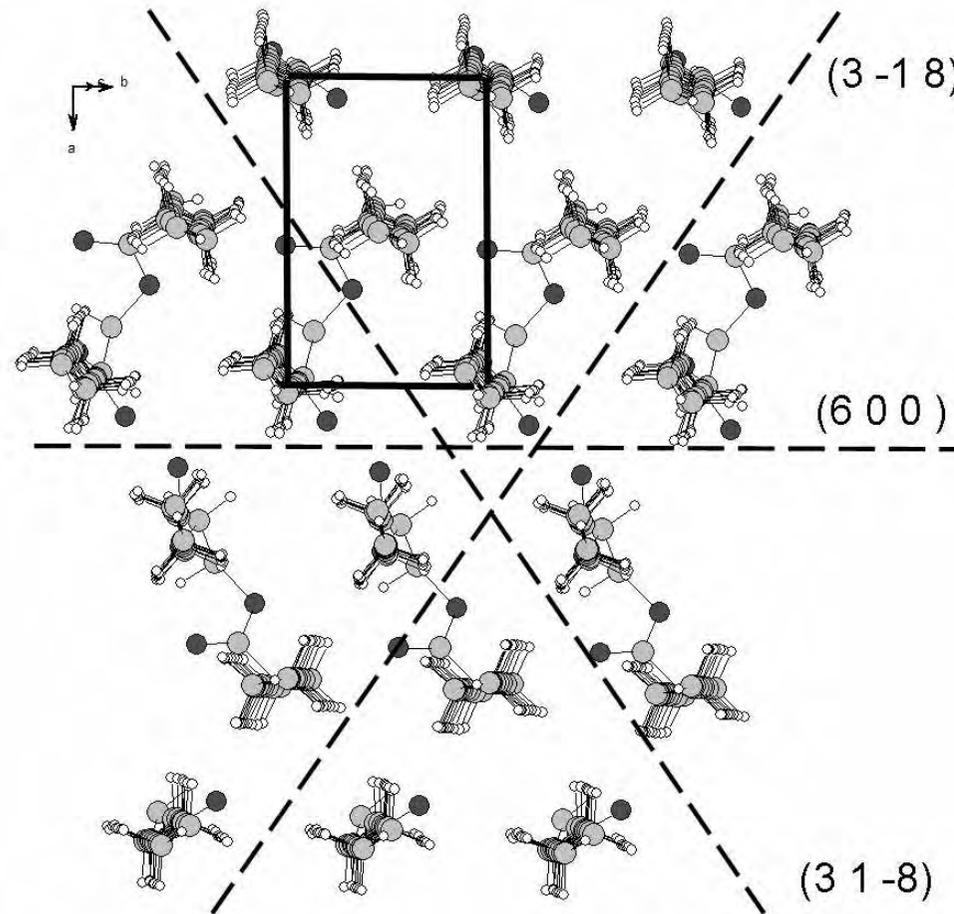
PSP (synchrotron data) rescaled to  $\text{Cu } K\alpha_1$

$(31\ 8)$   $(3\ 1\ -8)$  and  $(600)$  dominant

$Z = 8$

Van Mechelen et al. (2008) Acta Cryst B64, 249-259.

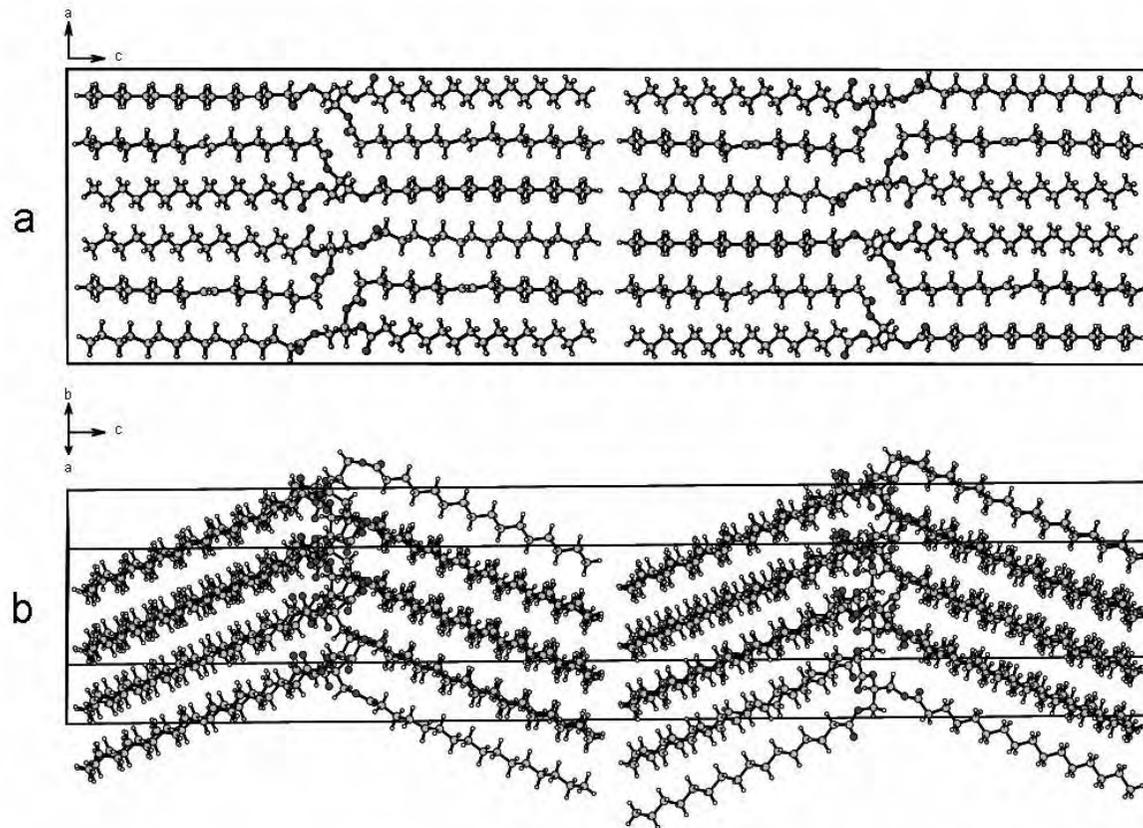
# Dominant reflections in $\beta'_{1-2}$ polymorphs (PSP)



Orthorhombic  $O\perp$  subcell

# Packing characteristics of the $\beta'_1$ -2 polymorphs

- Two pack layers
- Chair-shaped molecules, seat-facing, seat of chair is S or E
- $\beta'$  bend ( $\sim 130^\circ$ ) between the back and back leg of the chair
- Symmetric TAGs (PEP, PSP) have a [1-2] conformation



123

[1-2]  $\beta'_1$ -2 PEP

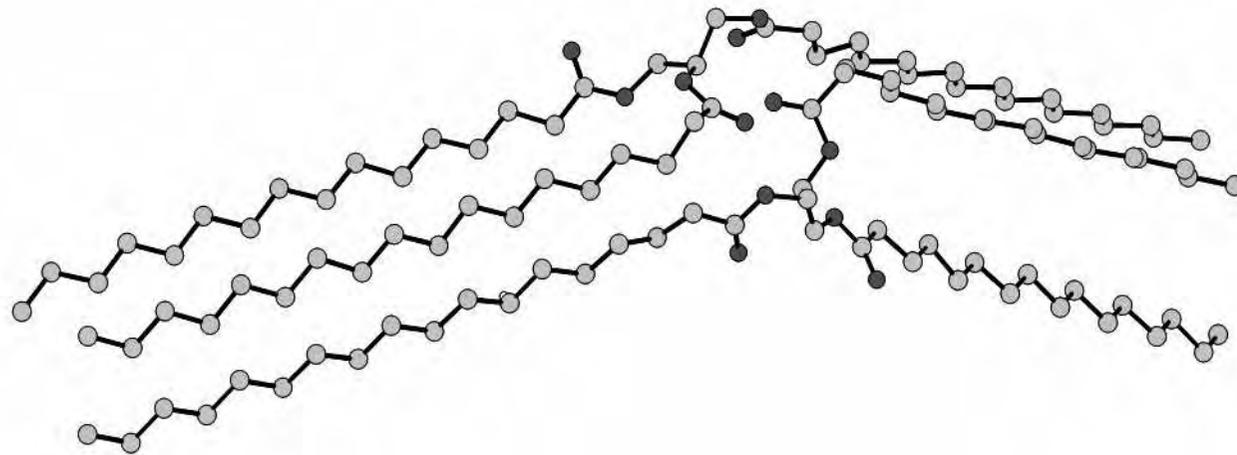
P3

P1 E

Bends in molecules point in the same direction

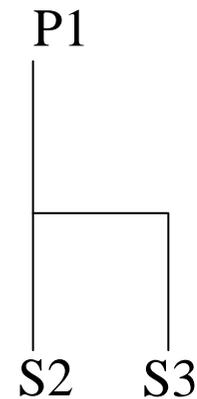
## Packing characteristics of the $\beta'_1$ -2 polymorphs

- Two pack layers
- Chair-shaped molecules, seat-facing, seat of chair is S or E
- $\beta'$  bend ( $\sim 130^\circ$ ) between the back and back leg of the chair
- Asymmetric TAGs (PPE, PPS) have a [2-3] conformation



Pair of PSS molecules with facing seats

123  
[2-3]  $\beta'_1$ -2 PSS

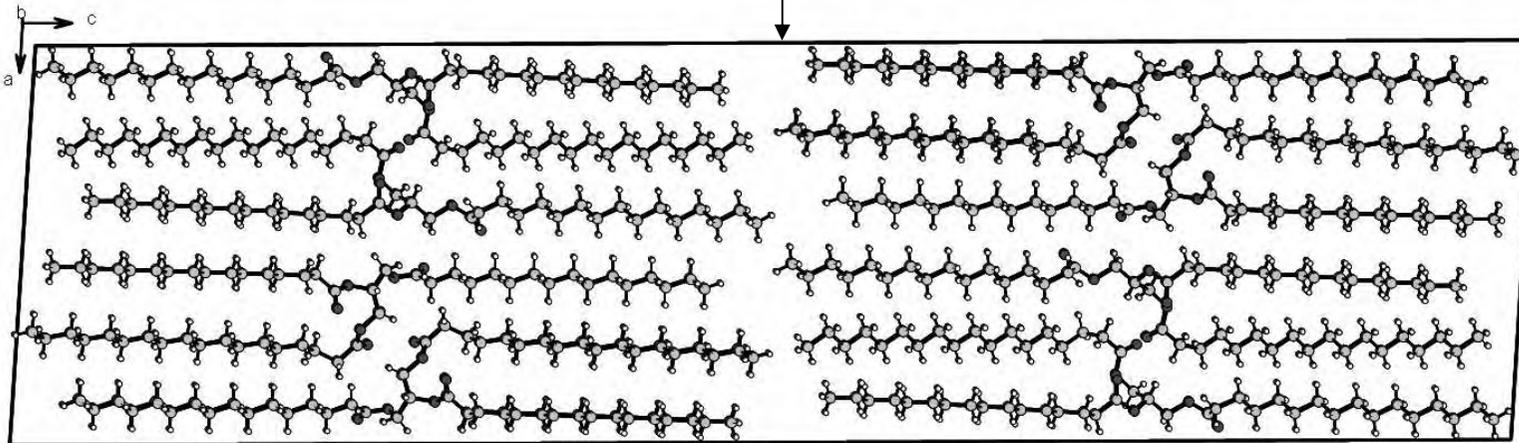


# Packing characteristics of the $\beta'_1$ -2 polymorphs

## Symmetric vs asymmetric TAGs

Methyl-end plane

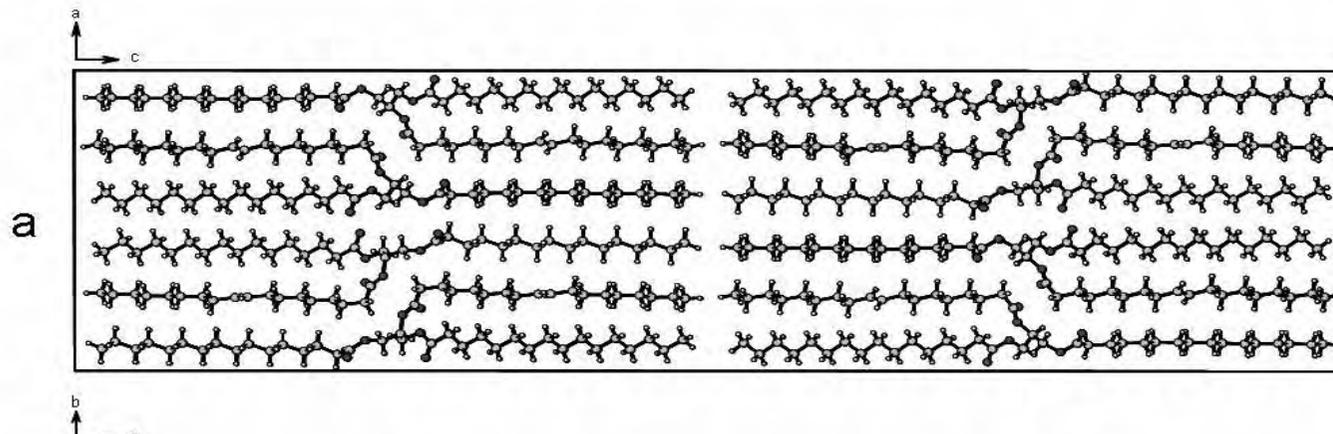
View parallel to  
b axis



Not aligned

[2-3]  $\beta'_1$ -2 PSS

$\alpha$  melt in  $\alpha \rightarrow \beta'_2$   
transition

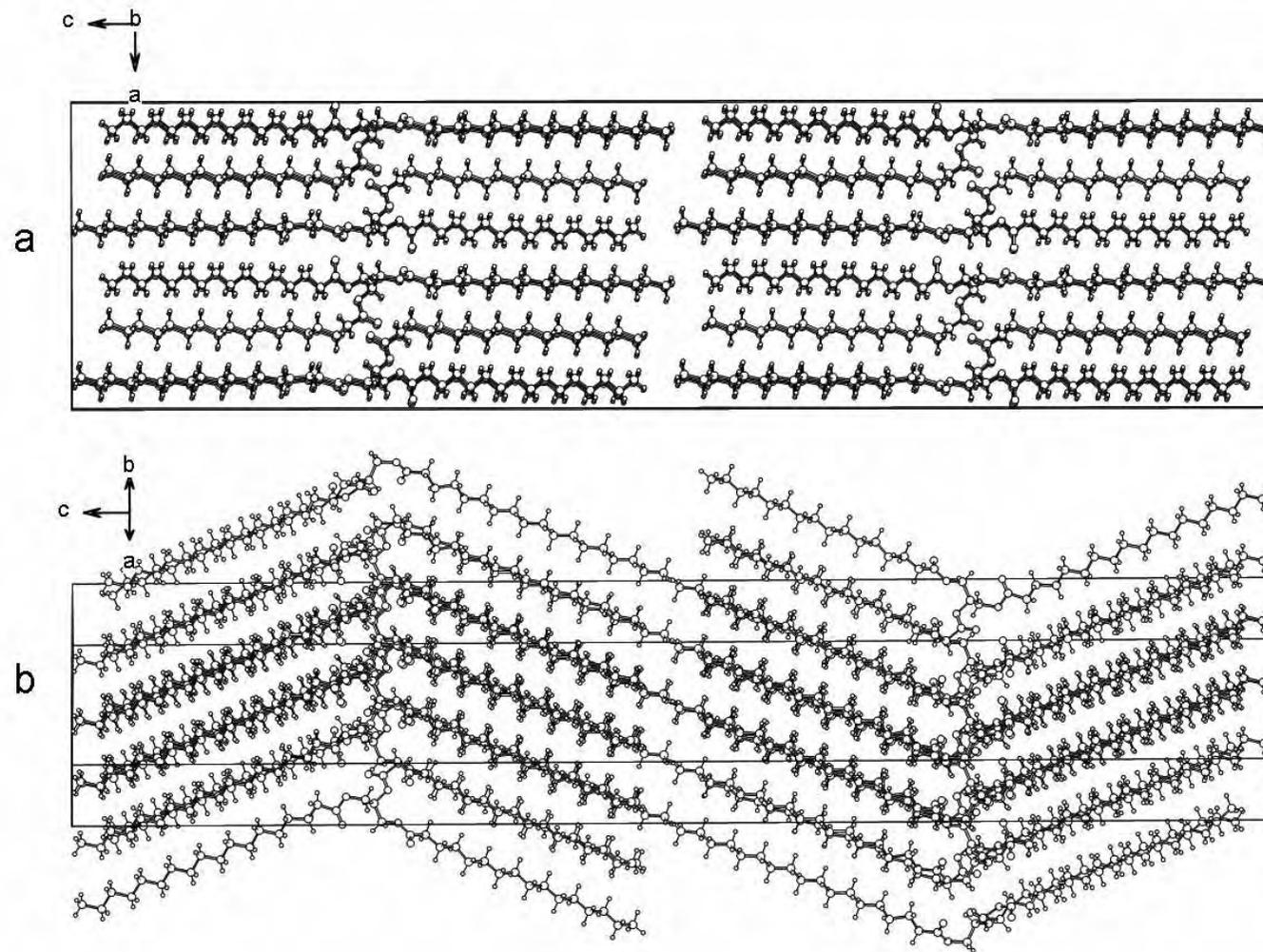


Aligned

[1-2]  $\beta'_1$ -2 PEP

No  $\alpha$  melt in  $\alpha \rightarrow \beta'_2$   
transition

# Packing characteristics of the novel $\beta'_0$ -2 PSS polymorph



[1-2] conformation

Two packs related by inversion centres

Stepped methyl end-plane

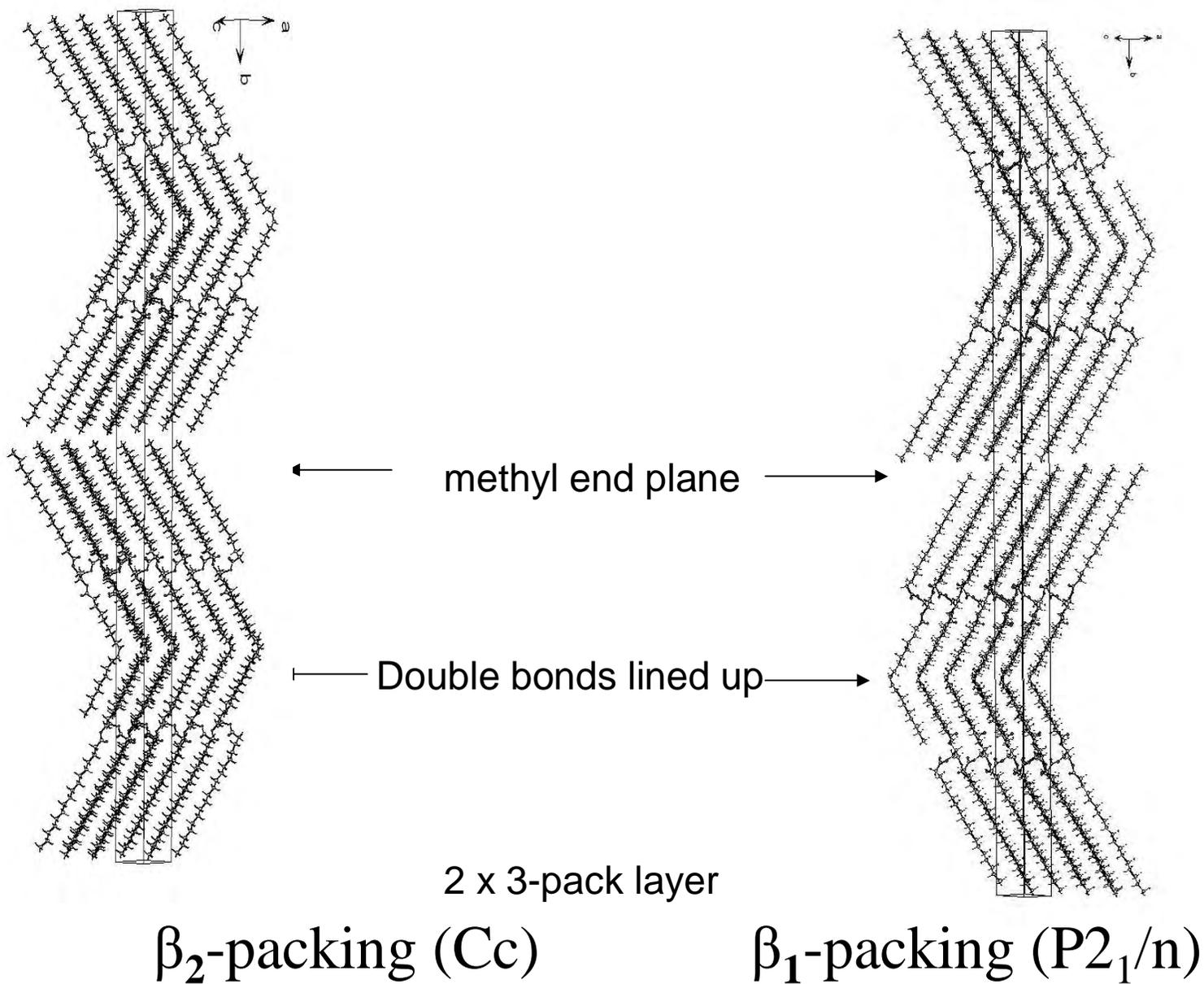
## $T_m$ (in K) and conformations of molecules in TAG polymorphs

Space group	I2	C2/c	P-1	
<b>TAG</b>	$\beta'_1-2$	$\beta'_0-2$	$\beta-2$	
PEP	329 [1-2]	-	327 [1-3]	} $\beta'$ stable
PSP	343 [1-2]	-	339.5 [1-3]	
PPE	320 [2-3]	-	320 [1-3]	} $\beta$ stable
PPS	332 [2-3]	-	338 [1-3]	
PSS	337 [2-3]?	339 [1-2]	339 [1-3]	

$\beta'_1-2$  [2-3]  $\Rightarrow$   $\beta'_0-2$  [1-2] would require an orientation inversion of every other two pack (similar as in  $\beta_2-3$  SOS  $\Rightarrow$   $\beta_1-3$  SOS)

# Three-pack conversion in SOS

Van Mechelen et al. (2006). *Acta Cryst. B* **62**, 1131-1138.



# Conclusions

## Time-resolved XRPD

- No definite influence of E vs. S except that E stabilizes  $\beta'_{2-2}$
- The  $\beta'_{2-2}$  is a less crystalline form of the  $\beta'_{1-2}$

## Structure determination from XRPD

- Symmetric and asymmetric  $\beta-2$  polymorphs are in the [1-3] conformation. No definite influence of E vs S.
- Symmetric  $\beta'_{1-2}$  (PSP, PEP) are in a [1-2] conformation,
- Asymmetric  $\beta'_{1-2}$  (PPS, PPE) in a [2-3] conformation.
- The  $\beta'_{0-2}$  PSS is in the [1-2] conformation

Solid state  $\beta'$  to  $\beta-2$  phase transition unlikely because of the different conformations

Low resolution data

$d > 3.0 \text{ \AA}$



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- V. Favre-Nicolin (CEA / Univ. J. Fourier)
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- The members of the User Committee of this project are thanked for stimulations discussions and continuous interest.

## References

- Van Mechelen, J.B. (2008). Triacylglycerol structures and the chocolate fat bloom mechanism. PhD Thesis. Universiteit van Amsterdam.
- Van Mechelen, J.B., Peschar, R. and Schenk, H. (2008) Structures of mono-unsaturated triacylglycerols. III. The  $\beta$ -2 polymorphs of *trans*-monounsaturated triacylglycerols, and related fully saturated triacylglycerols. (2008). Acta Cryst B64, 240-248.
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- Van Mechelen, J.B., Peschar, R. and Schenk, H. (2007) Progress in solving crystal structures of triglycerides from powder data using direct space techniques. In *IUCr Commission on Powder Diffraction. Newsletter 35, June* p41-44.  
*Electronic:*  
[http://www.iucr-cpd.org/PDFs/CPD\\_35\\_total.pdf](http://www.iucr-cpd.org/PDFs/CPD_35_total.pdf)

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